



U.S. Department of Energy



# Drip Shield Creep and Environmental Degradation Update

**Presented to:**  
**Nuclear Waste Technical Review Board**

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**Waste Package Modeling and Testing**

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**November 8-9, 2005**  
**Las Vegas, Nevada**

# Objectives

- **Describe drip shield materials selection basis**
- **Review**
  - **Environmentally induced cracking susceptibility of drip shield titanium alloys**
  - **Effect of creep on drip shield performance**
  - **General, localized and galvanic corrosion of drip shield materials**
- **Conclusions**



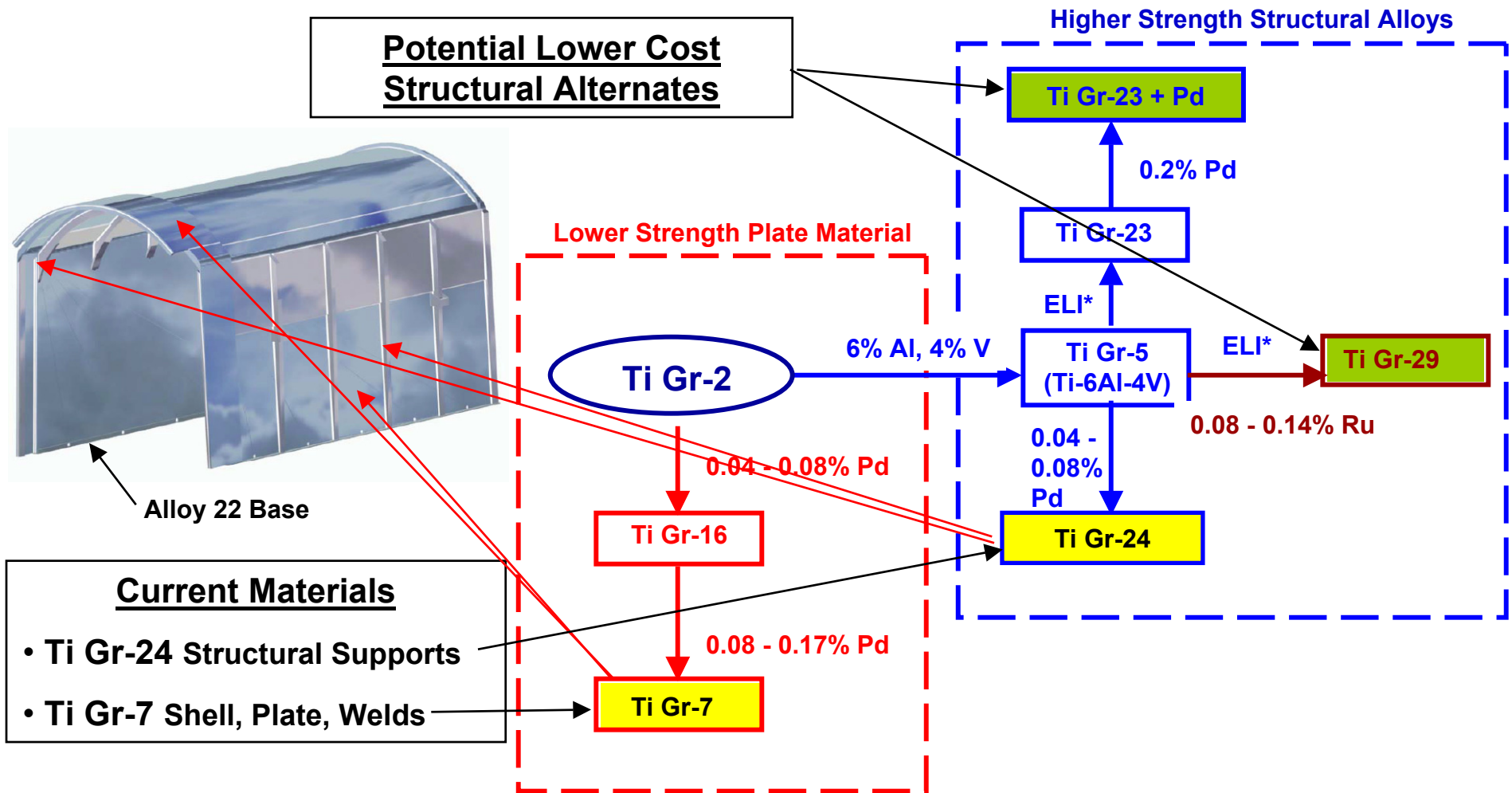
# Selection of Titanium for Drip Shield (DS)

- **DS materials selection based on series of YMP sponsored materials experts meetings & evaluations\***
  - **Selection of corrosion resistant materials narrowed to Alloy 22 for WP and Ti Gr-7 or Gr-16 for DS in 1999**
    - » Selection based on expected corrosion performance, defense in depth, fabrication ease, handling issues, cost impact and estimated lifetime
- **Ti Gr-24 (or Gr-29) selected later for DS structural supports**
  - **Literature indicates excellent general and localized corrosion performance**
- **Titanium alloys evaluated on other repository programs (including Canadian, Japanese and German programs)**

\* 1) Peer Review Report on Selection Criteria for the Yucca Mountain Project Waste Package Container Material, December 14, 1998,  
2) Selection of Candidate Container Materials for the Conceptual Waste Package Design for a Potential High Level Nuclear Waste Repository at Yucca Mountain, UCRL-ID-112-58, 1993,  
3) Waste Package Materials Selection Analysis, BBA000000-01717-0200-00020 REV 01, Table 7.3.1.2-1, 1997,  
4) Waste Package Containment Barrier Materials and Drip Shield Selection Report”, B00000000-01717-2200-00225 REV 00, 1999



# Current and Potential Alternate DS Materials



**Small Pd or Ru additions greatly enhance corrosion resistance**

\*ELI = Extra Low Interstitial (lower Fe, O and N gives higher toughness)



# Environmentally Induced Cracking Susceptibility of DS Alloys



# Environmental Cracking of DS Alloys

- DS alloys highly resistant but may not be immune to Hydrogen Induced Cracking (HIC) and Stress Corrosion Cracking (SCC)
- Prior relevant testing includes:
  - German\* slow strain rate tests (SSRT) of Ti Gr-7 in 5 M NaCl brine at 170°C
    - ◆ Concluded: Ti Gr-7 base metal and welds not sensitive to SCC
  - NRC funded SSRT of Ti G- 2, Gr-7 and Gr-5 in deaerated 95°C, 1 M NaCl with and without 0.1 M NaF \*\*
    - ◆ No SCC (or HIC) in notched specimens without 0.1 M NaF
    - ◆ With 0.1 M NaF, potential HIC observed for all three alloys
      - » Ti Gr-5 (basis for Ti Gr-24/29) more susceptible than Ti Gr-7
    - ◆ Aerated repository conditions and extended period of dry oxidation will provide HIC margin and resistance to fluoride effects

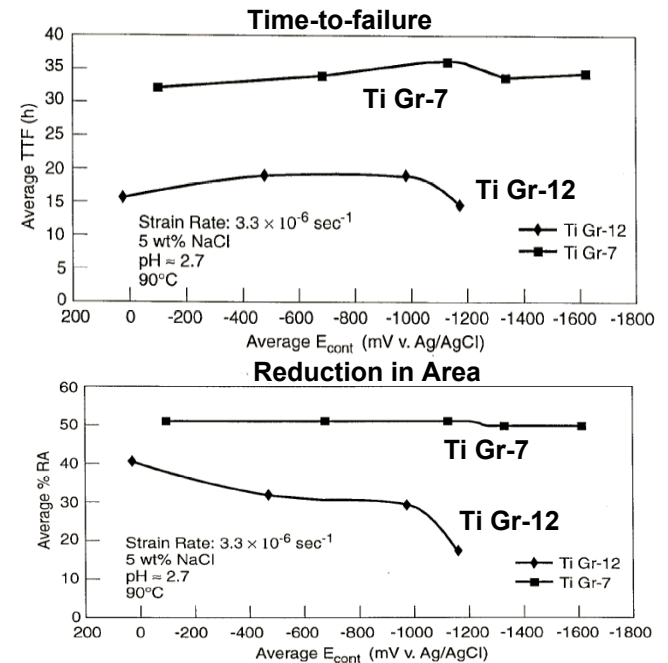
\* Smailos et al, Enresa final report, EC-Contract NO. F14W-CT95-0002, 1999

\*\* Pan et al., CNWRA 2003-02, Section 4.2

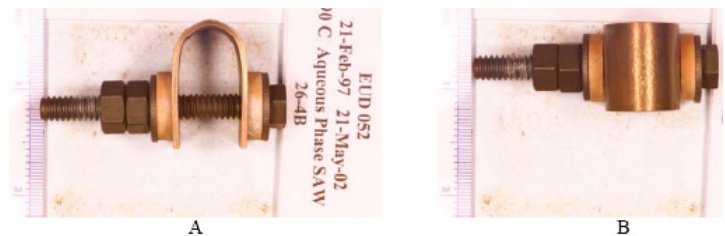


# Environmental Cracking of DS Alloys

- YMP SSRT\* on cathodically polarized Ti Gr-7 and Gr-12 (Gr-2 + .7 Ni, .3 Mo) in 90°C, 5 wt % NaCl, pH ~2.7
  - Times-to-failure (T-T-F) and ductility for Ti Gr 7 not significantly influenced by very negative potentials
    - ◆ Highly resistant to HIC
  - Ti Gr-12 showed significant drop in ductility and lower T-T-F at very negative potentials indicative of HIC.
- YMP U-bends of Ti Gr-7, 16 and 12 exposed in LLNL brines for 0.5-5.5 years\*\*
  - No SCC in Ti Gr-7, Gr-16 in 60-90°C SDW, SAW & SCW brines\*\*\* (0 to ~0.1 M F, pH ~3 to 10)
  - SCC in as-welded Ti Gr-12 90°C SCW (contains ~ 0.1 M F)



Typical Un-Cracked Ti U-bend Specimen after 5 years Exposure to SAW at 90°C



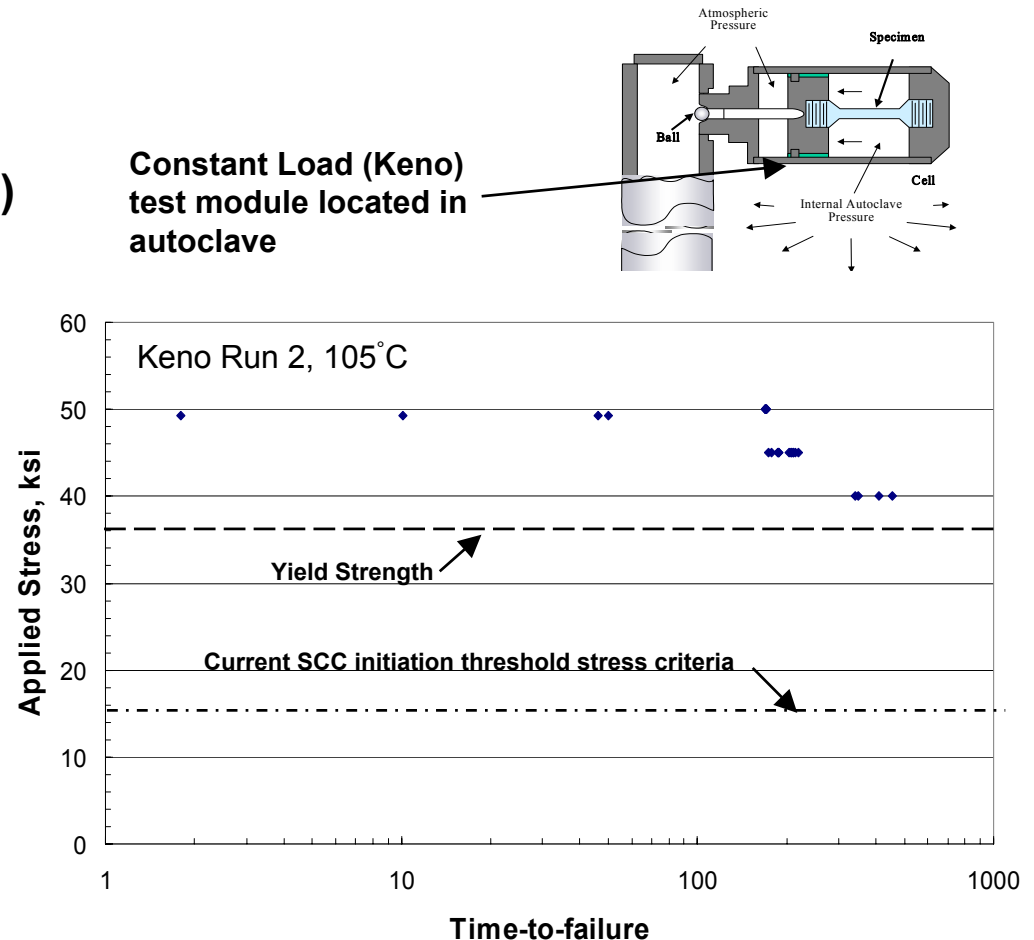
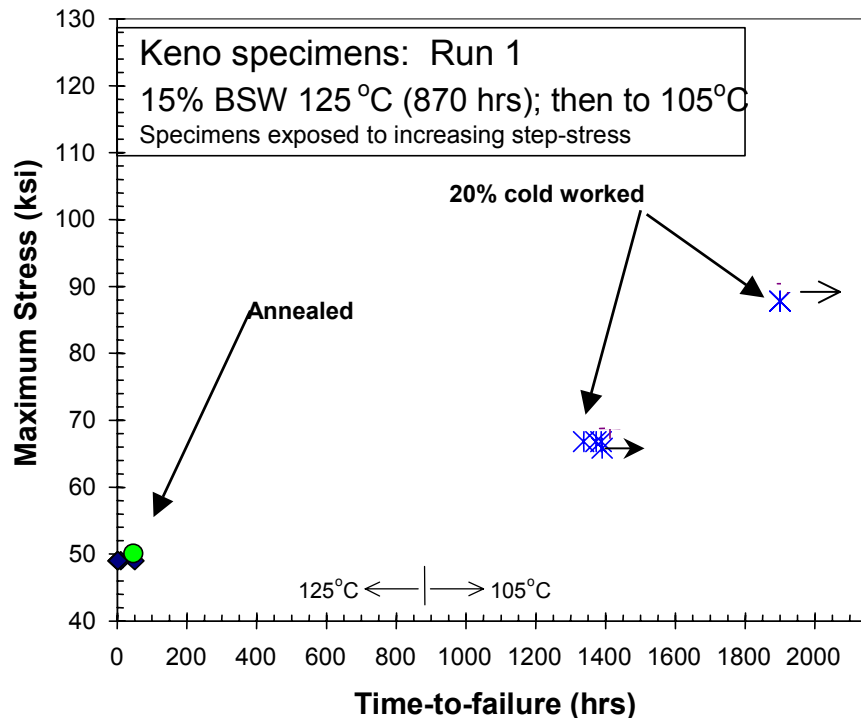
Exposed in LLNL Long Term Corrosion Test Facility (LTCTF)

\*Roy et al., Corrosion 2000, Paper 00188, \*\*Fix et al Paper 04551, Corrosion 04, \*\*\*SAW = Simulated Acidified Water, SCW = Simulated Concentrated Water, SDW = Simulated Dilute Water



# Environmental Cracking of DS Alloys

- **Two GE constant load Ti Gr-7 SCC initiation test campaigns\***
  - **15% BSW (~7500X J-13, pH ~12)**



- Failure in annealed Ti Gr-7 at 125°C at  $\geq 48$  ksi (~1.4YS)
- No cracking in 20%-Cold Worked Ti Gr-7 at 88 ksi (~1.2 YS)
- Annealed specimens fail at over-yield stresses in ~2-500 hours
- Tests at lower applied stress levels being initiated

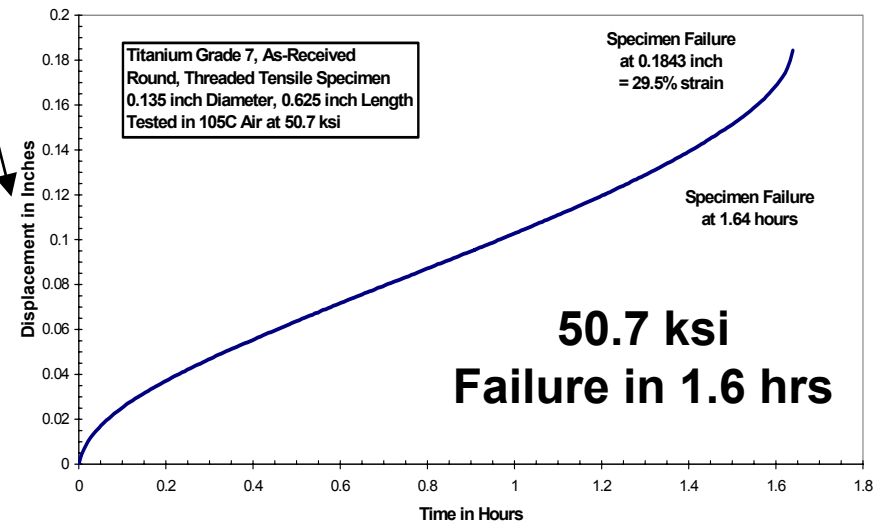
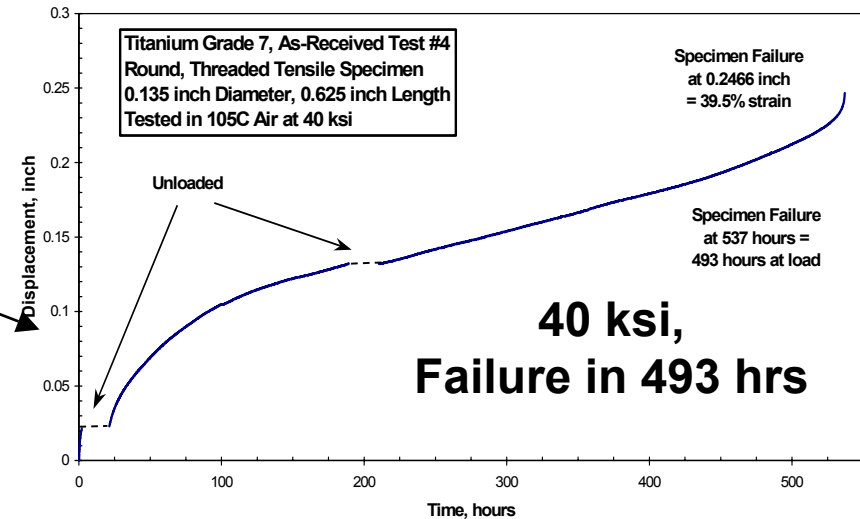
\* Young, et al, Proceedings, 11th Int'l Conference on Environmental Degradation, Paper No. 72115, 2003





# Comparison of Failure Times in SCC Tests with Air Creep Rupture Results

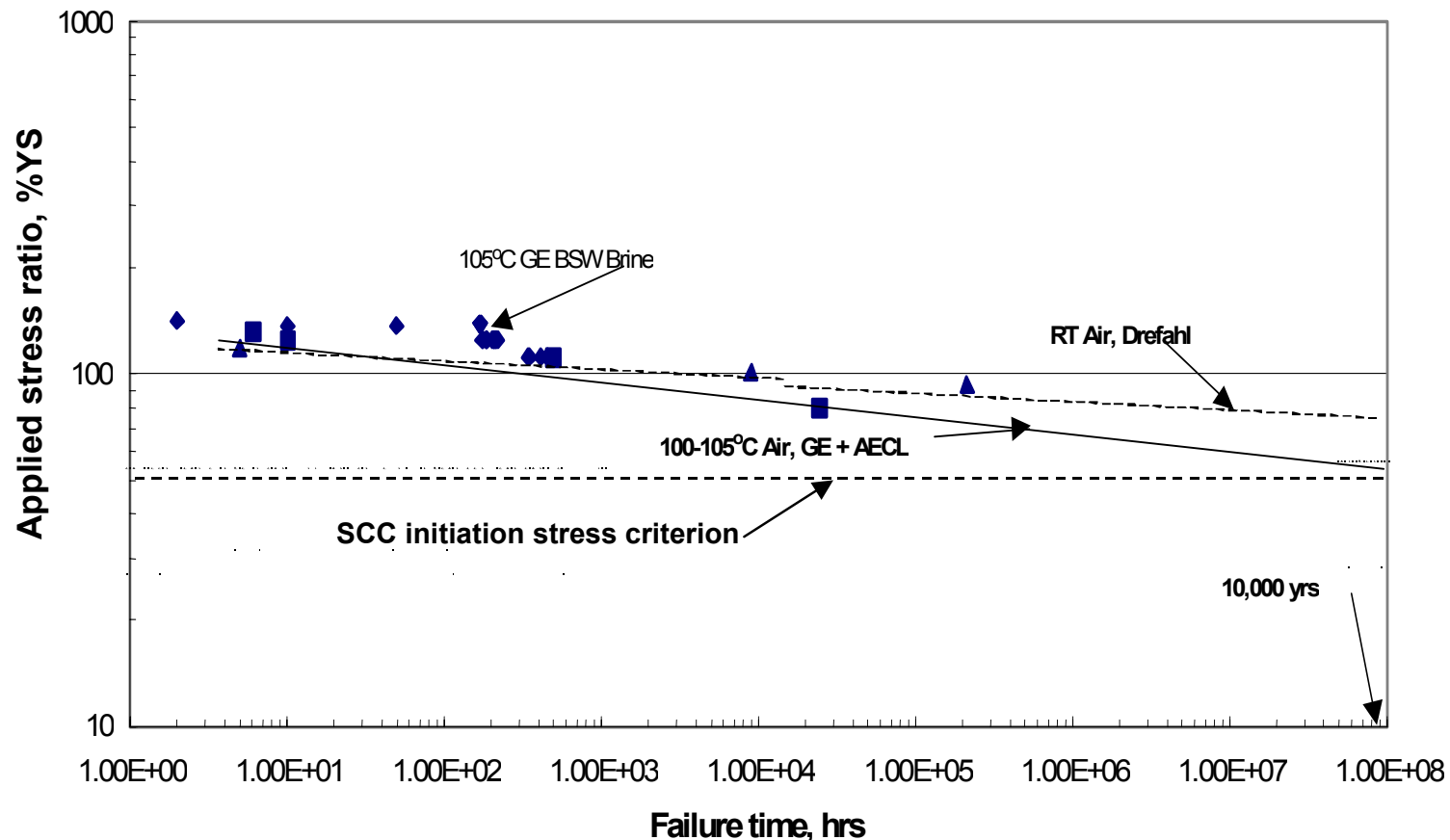
- Because of short brine failure times, air creep tests run
  - Typical Ti Gr 7 creep ruptures in air at 105°C at  $\geq 110\%$  YS\*
  - Other 100°C air tests\*\* indicate creep at  $\geq 60\%$  YS
- Specimen failure times in brine and air overlap at similar stresses
- Necessary to determine if failures due to SCC or creep rupture



\*Andresen et al., 12th Int'l Conference on Environmental Degradation, Salt Lake City, August 14-18, 2005, \*\*Dutton et al., Preliminary Analysis of the Creep Behaviour of Nuclear Fuel-Waste Container Materials, AECL-11495, COG-95-560-1, 1996



# Comparison of Constant Load Failure Times in 15% BSW Brine with Air Creep Rupture Results\*



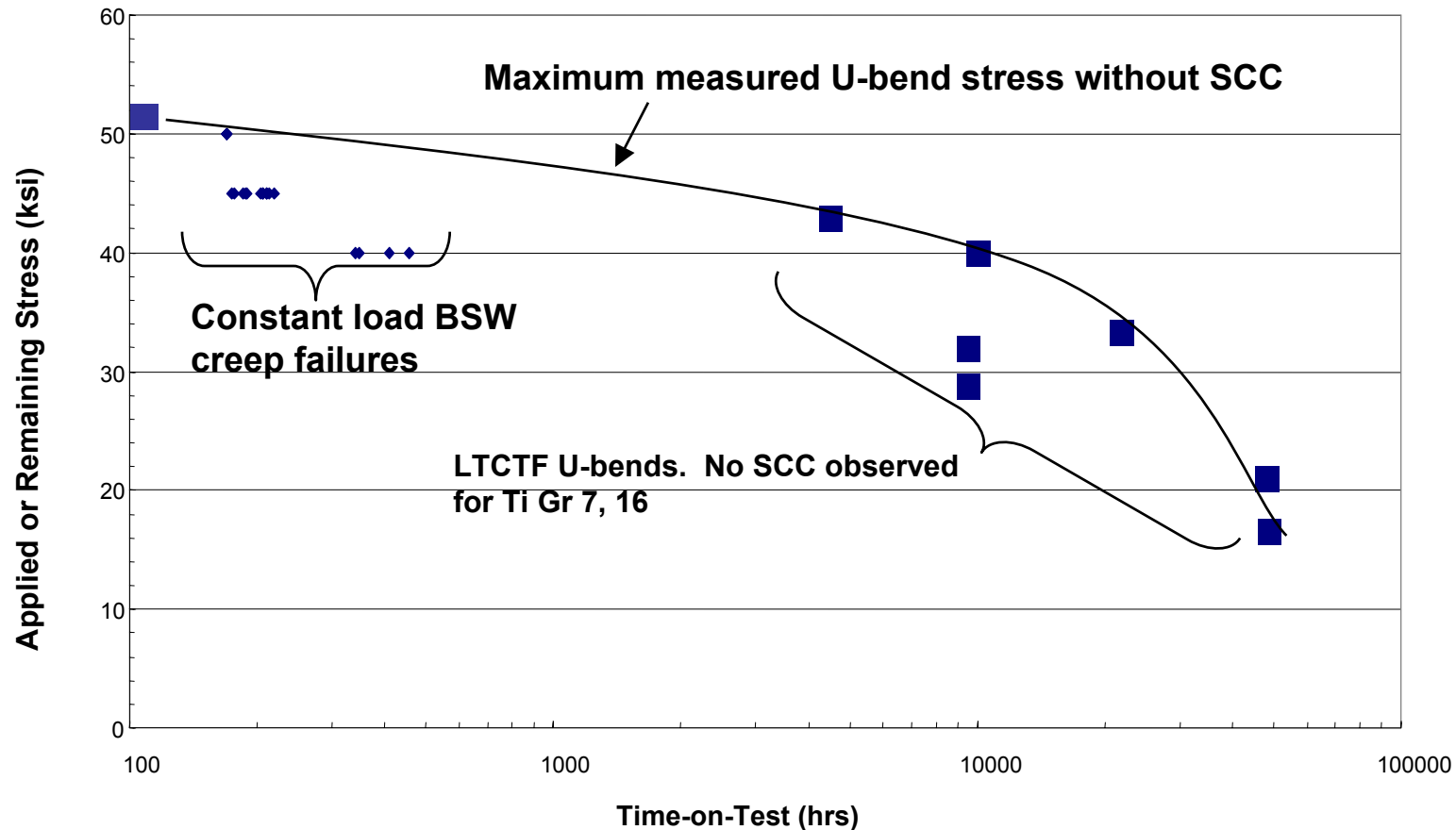
- **Constant load brine failures due to creep rupture rather than SCC**
- **Significant margin between minimum stress without SCC initiation and threshold stress criterion**

\*Andresen et al., 12th Int'l Conference on Environmental Degradation, Salt Lake City, August 14-18, 2005, Dutton et al., Preliminary Analysis of the Creep Behaviour of Nuclear Fuel-Waste Container Materials, AECL-11495, COG-95-560-1, 1996, Drefahl et al., Titanium Science and Technology (G. Lutjering, U.Zwicker and W. Bunk, editors), Deutsche Gesellschaft fur Metallkunde E.V., Germany.



# Stress Versus Time-on-Test in Brines

Constant Load Tests in 105°C, 15% BSW; U-Bend Test in 60-90°C SAW, SCW, SDW



- **Constant load specimens fail by creep rupture but U-bends exhibit no SCC initiation after 0.5-5.5 years exposure**
  - Maximum U-bend stresses (from X-ray diffraction)\* drop over time from creep relaxation
- **Stress relaxation beneficial in reducing rock fall impact residual stresses**

\*Lambda Research Report 1181-12611, MOL.20051031.0202

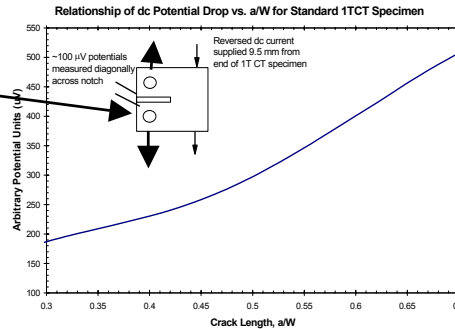


# Comparison of Crack Growth Rates in BSW and SCW Brines versus Air Creep Rate Results

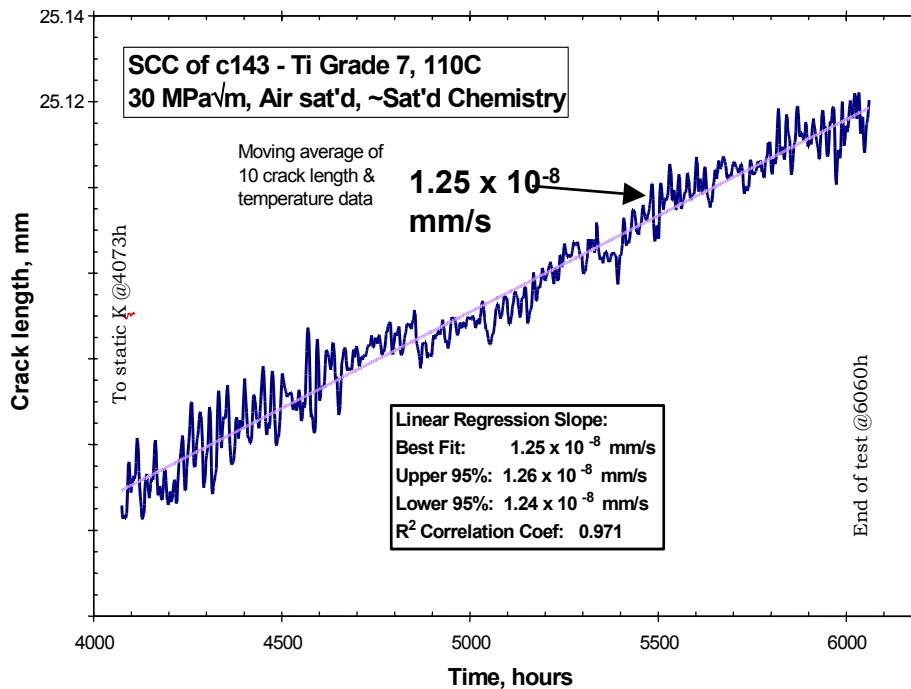


# Ti Gr-7 Crack Growth in 110°C in Air and BSW

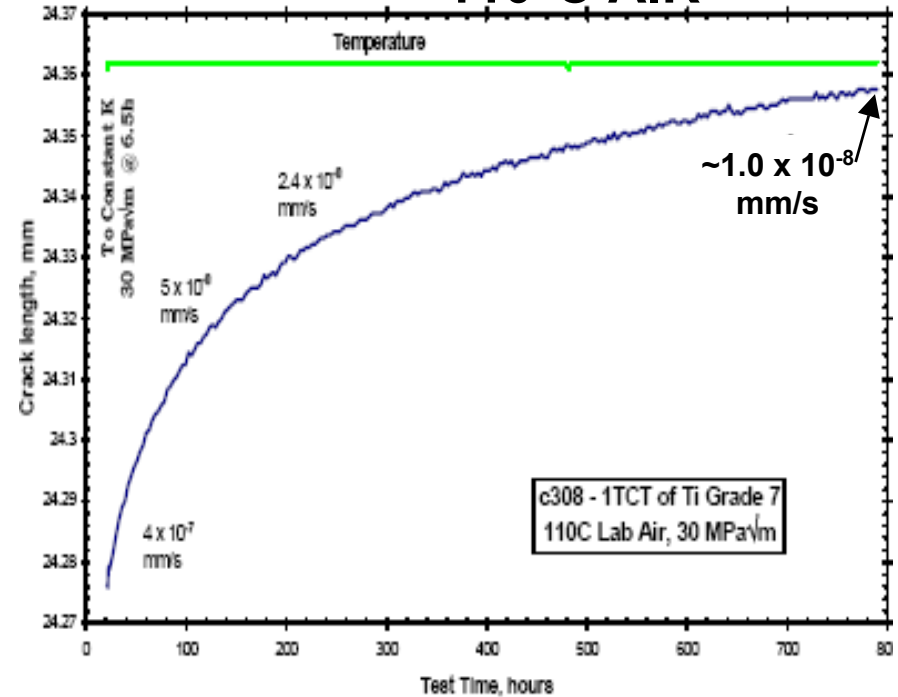
- Fracture mechanics type fatigue pre-cracked compact tension specimen
- Growth monitored using DC potential drop



## 110°C BSW



## 110°C AIR



- Constant load crack growth rates similar in air and BSW
- Consistent with significant creep component for Ti Gr-7 growth

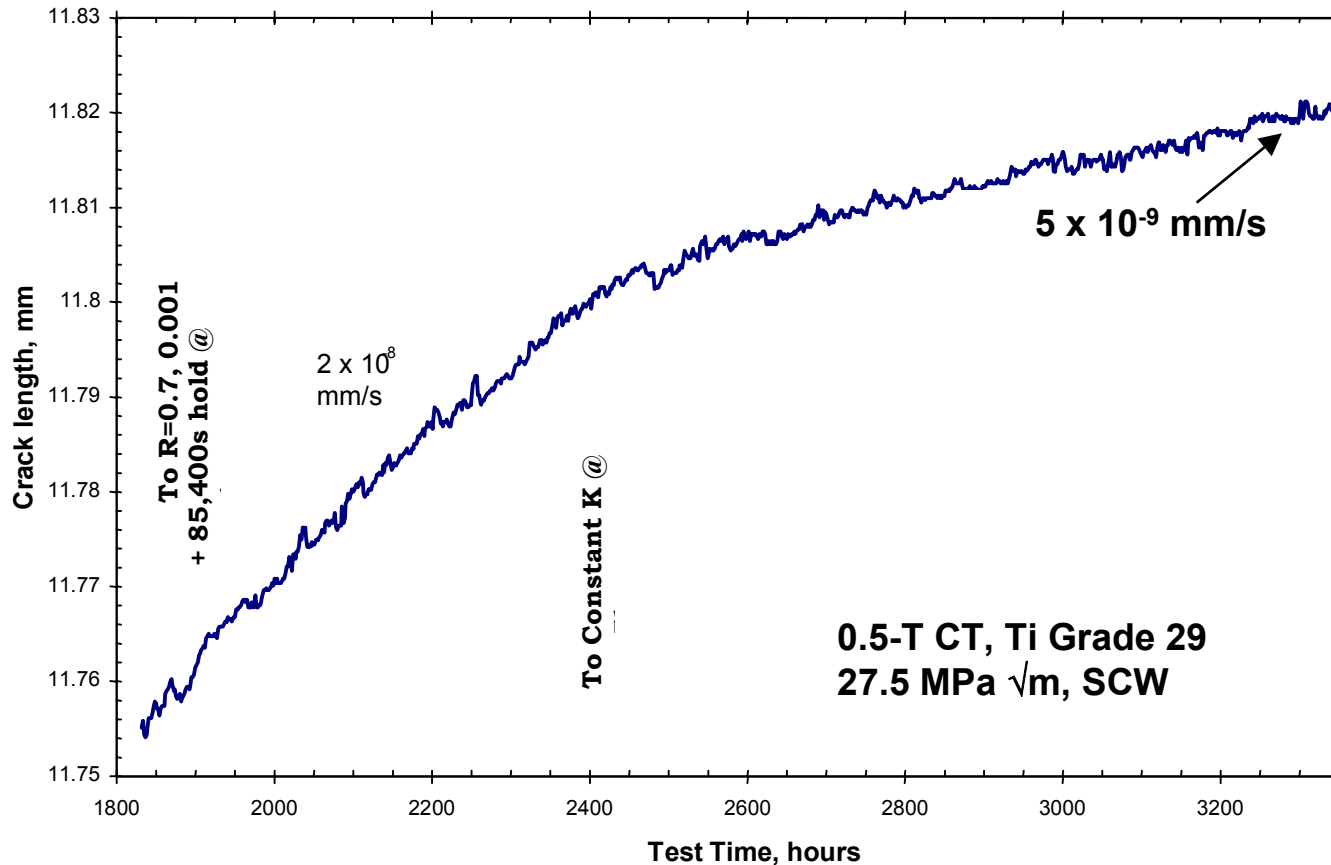
\*Andresen et al., 12th Int'l Conference on Environmental Degradation, Salt Lake City, August 14-18, 2005



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# Ti Grade 29 Crack Growth in 150°C SCW



- Ti Gr 29 (analog for Ti Gr 24) growth rate\* much higher than Ti Gr 7
- Since Ti Gr 29 significantly more creep resistant, crack growth likely due to SCC
  - Air control test planned to confirm crack growth mechanism

\*Andresen et al., 12th Int'l Conference on Environmental Degradation, Salt Lake City, August 14-18, 2005



# Effects of Creep on DS Performance



# Effects of Creep on DS Performance

- **As-emplaced DS resistant to creep and SCC**
  - Stress relief annealed condition
- **Two significant loading scenarios**
  - **Scenario 1: Rock rubble loads in lithophysal zones**
    - ◆ May lead to DS creep rupture and/or collapse
  - **Scenario 2: Seismic induced rockfall in non-lithophysal zones**
    - ◆ Residual stresses may initiate and propagate SCC
    - ◆ Seepage water pooling may increase flow rate through cracks
      - » However, cracks form at dent periphery not bottom of dent\*\*
- **Scenarios analyzed using literature benchmarked and conservative creep power laws\***
  - **Calculations\* indicate DS performance acceptable**



\*Creep Deformation of the Drip Shield, CAL-WIS-AC-000004 REV 0, \*\*ANL-WIS-PA-000002 REV 05, FEP 2.1.03.10.0B



# Effects of Creep on DS Performance

## Rock Rubble Loading

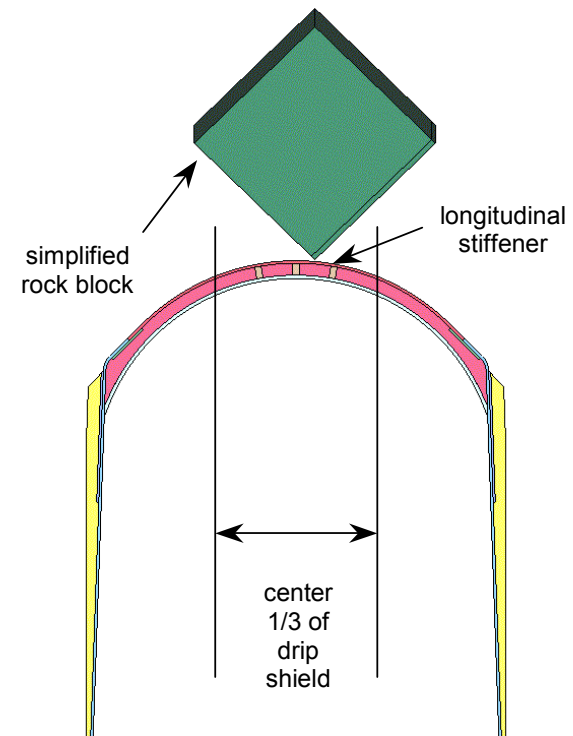
- **Maximum calculated creep strains less than 5%**
  - Acceptable limit is 10%
    - ◆ ~Onset of tertiary creep

## Rock Fall Impact

- **Residual stresses >50% YS**
  - Through-wall SCC can occur
  - Creep/stress relaxation calculation indicates SCC unlikely
    - ◆ Stresses relax below 50% of YS in less than 100 years at 30°C
    - ◆ Stresses relax below 65% of YS in less than 10 years at 150°C
    - ◆ Consistent with measured U-bend stress versus time results

*Maximum creep strains 10,000 years after emplacement*

Realization	Creep Strain (%)
1	4.20
2	1.74
3	2.01
4	3.11
5	1.28
6	2.95



# General, Localized and Galvanic Corrosion of DS Alloys



# Drip Shield General Corrosion

- **Ti-Gr 7 Plate Material\***

- One year Ti Gr-16 exposed specimens used for PA (Performance Assessment)

- ◆ Corrosion rate decreases with time\* – not used in PA
    - ◆ Ti-7 more resistant than Ti-16
    - ◆ Median value of <10 nm/yr after 2.5 years exposure

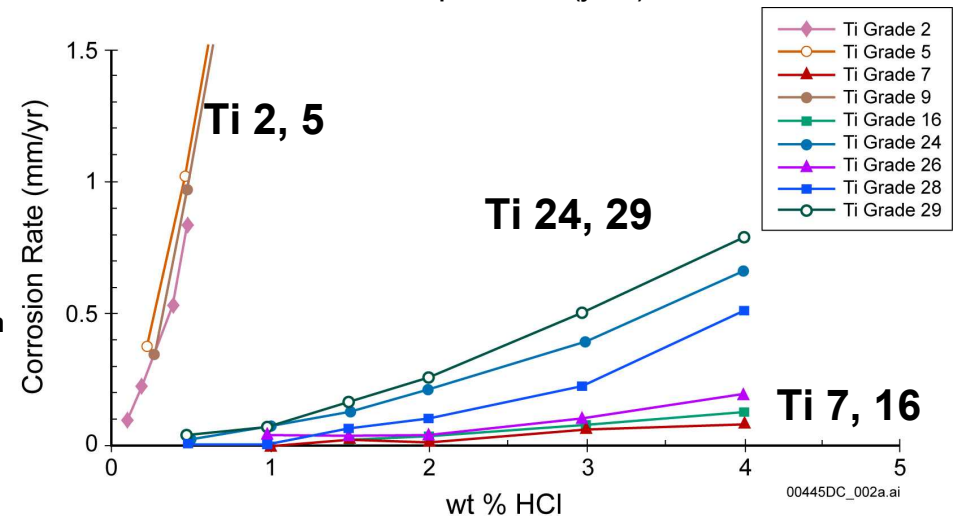
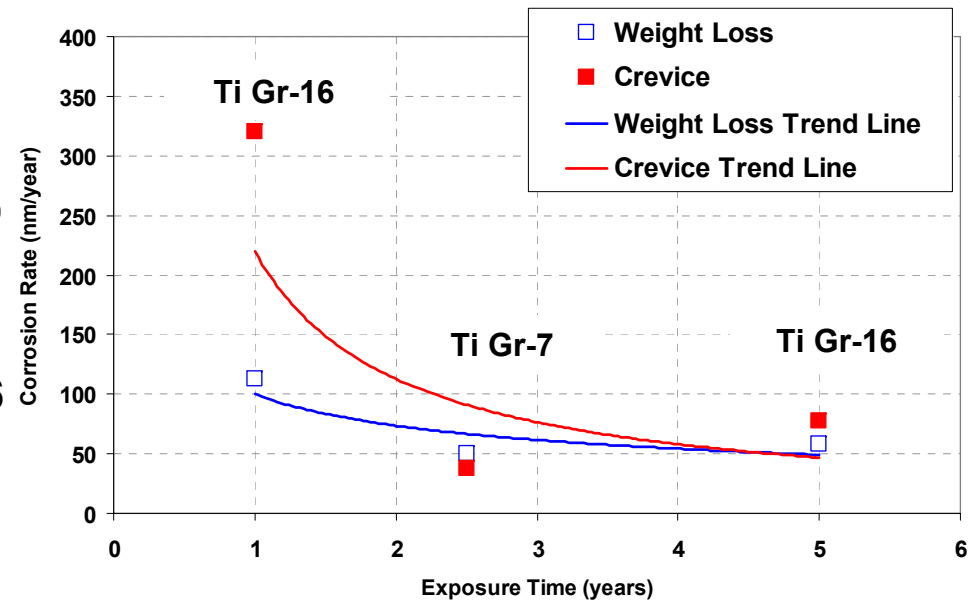
- **Ti-24/29 Supports\*\***

- Accelerated tests in acidic chloride media indicate GC rate is 4-5x Ti-7 GC Rate\*\*

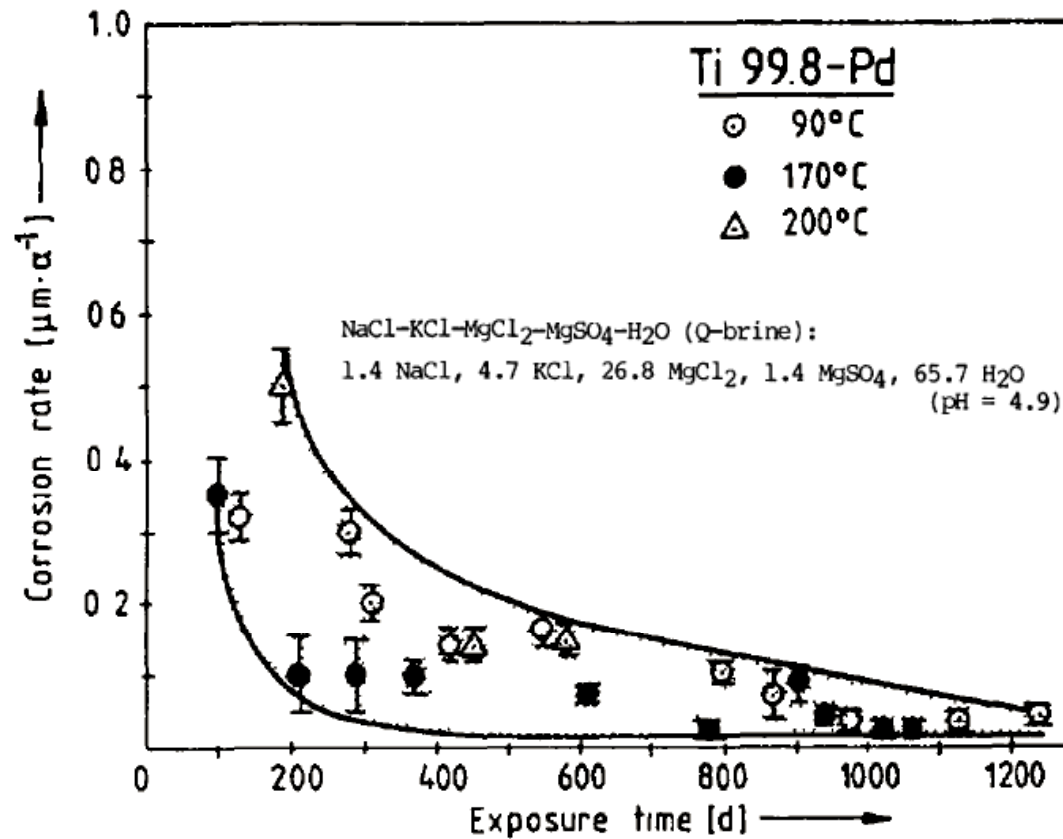
- ◆ Still extremely low

\*Hua et al., A Review of Titanium Grade 7 and Other Titanium Alloys in Nuclear Waste Repository Environments, Corrosion, Vol 61, No. 10, pp. 987-1003, 2005, ANL-EBS-MD-000004, REV 02; \*\*R.W. Schutz, Platinum Group Metal Additions to Titanium: A Highly Effective Strategy for Enhancing Corrosion Resistance, Corrosion Vol 59, No. 12, pp.1043-1057, 2003

Maximum Measured Corrosion Rates in LTCTF



# Ti Grade 7 General Corrosion in Q-Brine\*

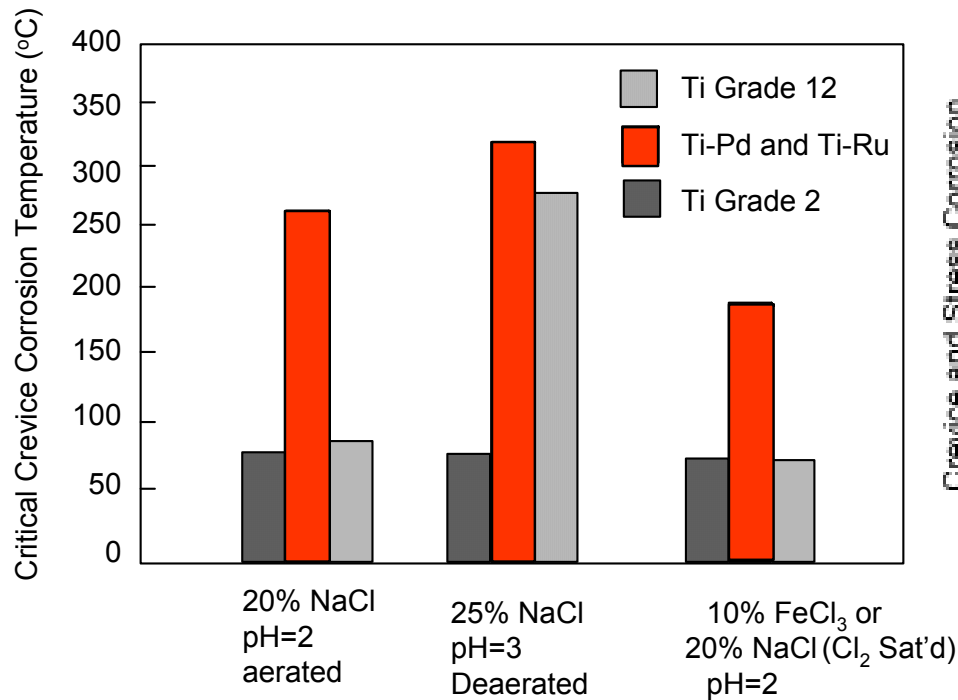


- General corrosion in Q-Brine consistent with LTCTF results
- Rate is temperature independent up to 200°C

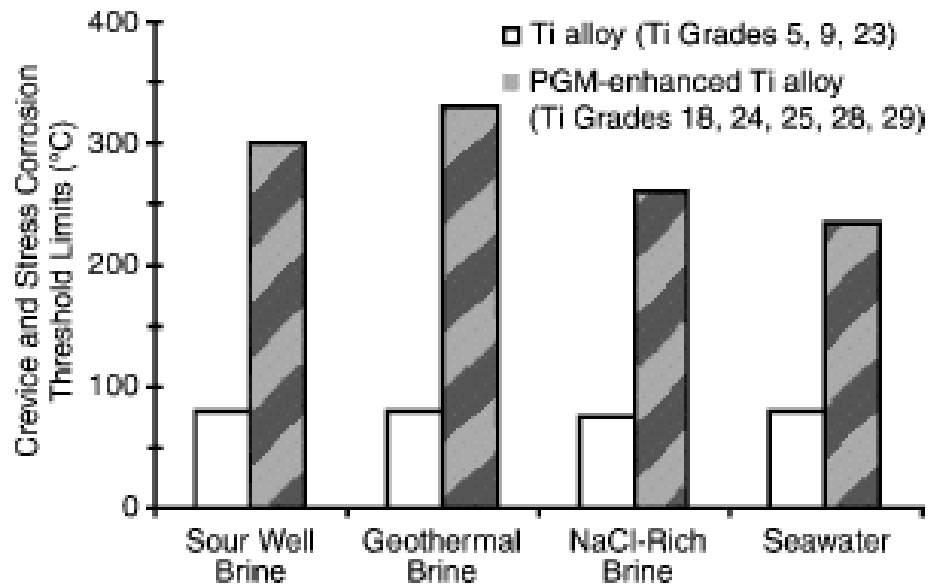
\* E. Smailos, R. Köster, Corrosion studies on selected packaging materials for disposal of high level wastes, **IAEA-TECDOC-421**, PROCEEDINGS OF A TECHNICAL COMMITTEE MEETING ON MATERIALS RELIABILITY IN THE BACK END OF THE NUCLEAR FUEL CYCLE ORGANIZED BY THE INTERNATIONAL ATOMIC ENERGY AGENCY AND HELD IN VIENNA, 2-5 SEPTEMBER 1986



# Critical Crevice Temperatures for Ti-Pd and Ti-Ru Containing Alloys



Crevice Ti-Gr 7, 12 and 16 in a range of aggressive acidic and highly oxidizing chloride environments\*



Crevice and stress corrosion temperature thresholds for alpha-beta alloys (e.g., Ti Gr-24 and 29) in aqueous chloride media\*\*

- **Ti-Gr 7, 24 and 29 highly resistant to crevice corrosion in chloride brines**

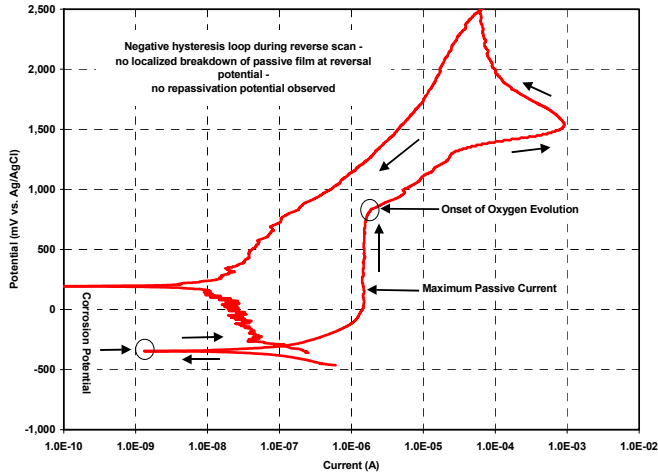
\*Schutz, R.W., "Ruthenium Enhanced Titanium Alloys: Minor Ruthenium Additions Produce Cost Effective Corrosion Resistant Commercial Titanium Alloys, "Platinum Metals Review, 40 (2), 54-61, 1996

\*\*Schutz, R.W., "Platinum Group Metal Additions to Titanium: A Highly Effective Strategy for Enhancing Corrosion Resistance", Corrosion, Vol 59, No. 12, pp. 1043-1057, 2003

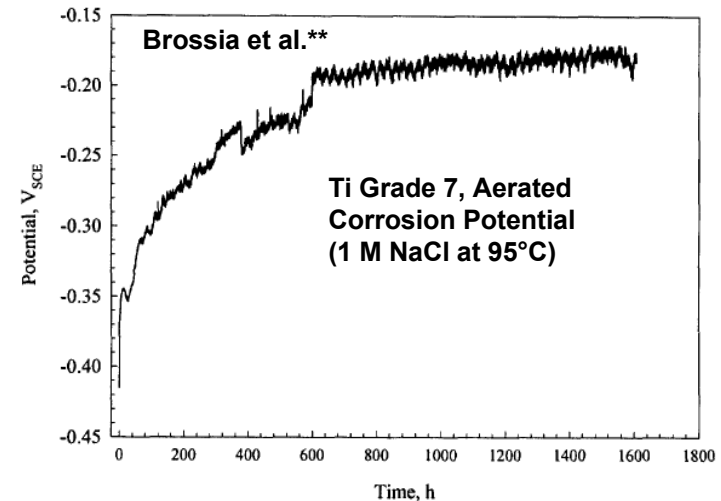
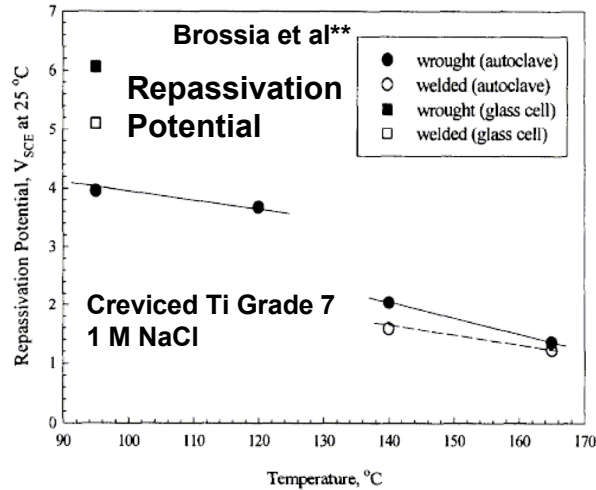
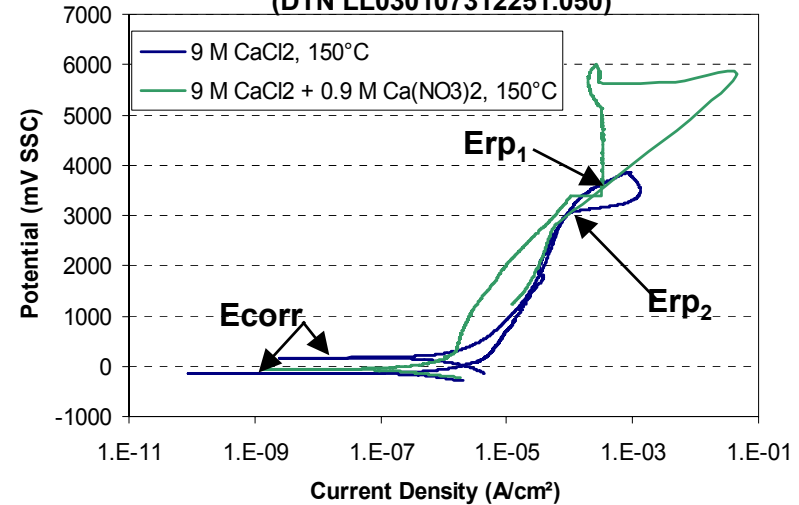


# Localized Corrosion of Ti Gr-7 at High Temperatures

120°C Simulated Saturated Water  
3.6 M KCl, 21.2 M NaNO<sub>3</sub>, NO<sub>3</sub>:Cl = 5.9\*



LLNL Cyclic Polarization  
(DTN LL030107312251.050)



- Localized Corrosion margin ( $\Delta E = E_{crit} - E_{corr}$ ) very large up to 165°C\*

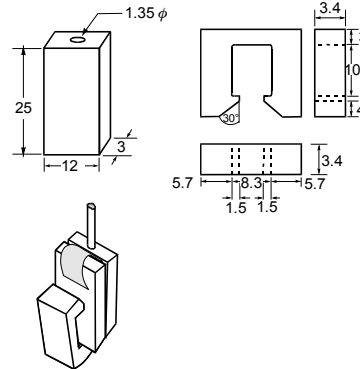
\*Hua et al., A Review of Titanium Grade 7 and Other Titanium Alloys in Nuclear Waste Repository Environments, Corrosion, Vol 61, No. 10, pp. 987-1003, 2005 \*\*Brossia et al., Paper NO. 00211, Corrosion 2000, NACE



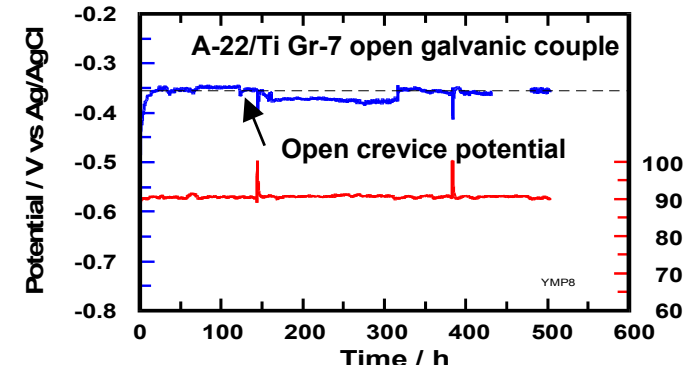
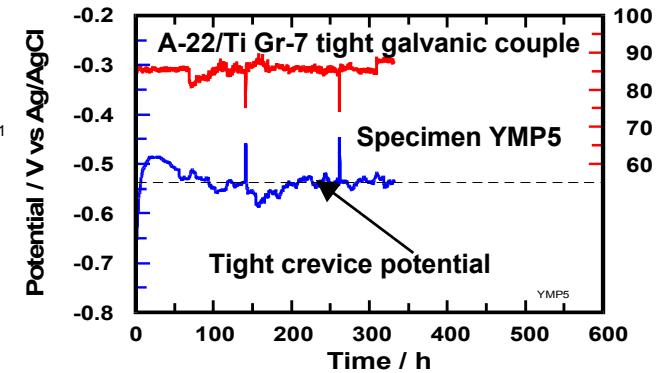
# Galvanic Corrosion of Ti Gr-7

## Ti Gr-7/Alloy 22 Galvanic Couples

- A series\* of galvanic couples tested in non-deaerated 90°C SCW
  - No evidence of galvanic or crevice corrosion
  - Tight crevice  $E_{corr}$  similar to uncreviced Ti Gr-7\*\*\*
  - Open crevice  $E_{corr}$  lies between Alloy 22 and Ti Gr-7

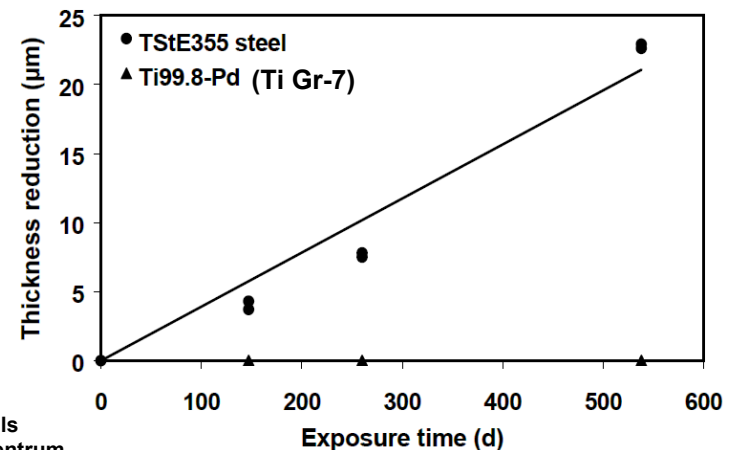


Creviced galvanic couple specimens



## Ti Gr-7/Carbon Steel Tight Contact Couples

- Ti Gr-7/carbon steel couples\*\* were exposed over 500 days in 150°C, 26% NaCl, pH = 6.5
  - Ti Gr-7 corrosion rate remains very low, < 0.07  $\mu\text{m}/\text{yr}$
  - Metallography indicated no evidence of corrosion attack



\*Ikeda et al., Corrosion of Dissimilar Metal Crevices in Simulated Concentrated Water Solutions at Elevated Temperatures, AECL-12167 REV 00, January 2003, \*\*Smialos et., Long-Term Performance of Candidate Materials for HLW/Spent Fuel Disposal Containers, Wissenschaftliche Berichte FZKA 6706. Fprscjingszentrum Karlsruhe, 2002, \*\*\*ANL-EBS-MD-000004, REV 02



# Conclusions

- **Titanium DS alloys highly resistant to SCC and HIC in repository relevant environments**
- **General corrosion rates extremely low and temperature independent to at least 200°C**
- **Large localized corrosion margins – LC not expected under repository relevant conditions**
- **Accelerated galvanic corrosion not expected for Ti-Gr-7 couples between Alloy 22 or carbon steel**
- **DS creep deformation can occur under rock rubble loading but calculation shows resulting strains acceptable**

